# Magnetic properties of YCo<sub>5</sub> (70%wt)+Y<sub>2</sub>Co<sub>17</sub> (30%wt) nanocomposite powders at low temperatures

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## Abstract

Nanostructured YCo<sub>5</sub> (70%wt)+Y<sub>2</sub>Co<sub>17</sub> (30%wt) composite powders were prepared by mechanical milling and subsequent annealing at 1073K for 1.5 min. The average grain size <D> of the YCo<sub>5</sub> and Y<sub>2</sub>Co<sub>17</sub> phases, obtained from XRD data, was 14 and 12 nm, respectively. The temperature dependence of the magnetic properties was studied by DC magnetization measurements at temperatures *T* ranging from 3 to 300 K. Hysteresis loops (H<sub>max</sub> = 70 kOe) show that both the coercivity H<sub>c</sub> and the squareness  $\sigma_r/\sigma_{max}$  are temperature dependent The coercivity increases from 12 kOe at room temperature to 18 kOe at T = 3K. The observed enhanced remanence ( $\sigma_r/\sigma_{max}$ >0.5) indicates that a strong exchange coupling is present at all temperatures used in this study. The maximum magnetization  $\sigma_{max}$  changes little with temperature and has a value of about 70% of the effective saturation magnetization of the title compound.

Keywords: Nanocrystalline material; Hard magnet; Nanostructured magnet; High coercivity; Permanent magnet.

## Introduction

The discovery of spring exchange coupling between hard and soft magnetic grains was made in 1991 when it was shown that a mixture of Nd<sub>2</sub>Fe<sub>14</sub>B and Fe<sub>3</sub>B or α-Fe could in principle lead to energy products (BH)<sub>max</sub> higher than those achieved in single-phase magnets [1]. Since then many theoretical and experimental investigations



have been carried out on various exchange-coupled nanocomposite materials with the goal of enhancing their magnetic properties [2–6]. Particularly, the hexagonal RECo<sub>5</sub> system (RE=rare earth) has been mixed with  $\alpha$ -Fe [7] and RE<sub>2</sub>Co<sub>17</sub> in order to improve the magnetization [8]. Besides the well-known temperature dependence of the RECo<sub>5</sub>-phase anisotropy constant, the exchange coupling between the soft and hard phases itself might be affected by the temperature. Consequently, it is important to investigate the temperature dependence of the magnetic properties of nanocomposite powders [9].

In this work we present the low-T behavior of the magnetic properties of  $YCo_5$  (70%wt)+Y<sub>2</sub>Co<sub>17</sub> (30%wt) nanocomposite powders prepared by mechanical milling and subsequent annealing at 1073K for 1.5 min, followed by quenching in water.

### Experimental

The starting materials were Y and Co ingots with purity of 99.9% and 99.8%, respectively. Alloys with nominal composition YCo<sub>5</sub> and Y<sub>2</sub>Co<sub>17</sub> were prepared by arc melting pure elements in an Ar atmosphere. The ingots were turned and re-melted four times to ensure homogeneity. The as-cast ingots were then coarsely pulverized and mixed to obtain 3 g of powders containing 70%wt of YCo<sub>5</sub> and 30%wt of Y<sub>2</sub>Co<sub>17</sub>. Afterwards, the powders were mechanically milled for 240 min under Ar atmosphere using a SPEX 8000 ball mill with a ratio of powders to balls of 1:8. The as-milled amorphous powders were annealed at 1073K for 1.5 min in high-vacuum vycor tubes, followed by quenching in water. More details about the sample preparation can be found elsewhere [10]. Structural analysis was carried out from X-ray diffraction (XRD) patterns collected on a Siemens D5000 diffractometer with Cu-K $\alpha$  radiation ( $\lambda$ =1.5406Å). The magnetic properties were measured on a Quantum Design physical



property measurement system (PPMS) with a maximum applied field,  $H_{max} = 70$  kOe. Microstructural observations were carried out with a JEOL 2100 F transmission electron microscope (TEM).

#### **Results and discussion**

Fig. 1 shows the XRD pattern of YCo<sub>5</sub> (70%wt)+ Y<sub>2</sub>Co<sub>17</sub> (30%wt) nanocomposite powders obtained after 4 h of mechanical milling and a subsequent annealing at 1073K for 1.5 min followed by quenching in water. The hexagonal CaCu<sub>5</sub>-type (PDF #17-078) and rhombohedral Th<sub>2</sub>Zn<sub>17</sub>-type (PDF #18-434) structures were used to index all the observed peaks of YCo<sub>5</sub> and Y<sub>2</sub>Co<sub>17</sub>, respectively. The full-width at half-maximum (FWHM) of the YCo<sub>5</sub> and Y<sub>2</sub>Co<sub>17</sub> peaks was used in the Scherrer equation [11] to independently estimate the average grain size <D> of the two phases that make the nanocomposite. Average grain sizes of 14 and 12nm were found for the YCo<sub>5</sub> and Y<sub>2</sub>Co<sub>17</sub> phases, respectively.

Fig. 2 shows the virgin magnetization curves and hysteresis loops of YCo<sub>5</sub> (70%wt)+Y<sub>2</sub>Co<sub>17</sub> (30%wt) nanocomposite powders annealed at 1073K for 1.5 min measured at two different temperatures, 295 and 3K. Both virgin curves indicate the same pinning-type magnetization mechanism as the one reported for single-phase YCo<sub>5</sub> nanostructured powders [12–14]. Magnetic saturation is not observed for the maximum applied field (H<sub>max</sub>=70 - kOe) at any of the two temperatures, as expected for nanostructured magnetic materials [15]. In addition, the maximum magnetization  $\sigma_{max}$  values of the YCo<sub>5</sub> (70%wt)+Y<sub>2</sub>Co<sub>17</sub> (30%wt) nanocomposite at 3 and 295K are almost the same.





Fig. 1. X-ray diffraction pattern of (70%)  $YCo_5 + (30\%) Y_2Co_{17}$  annealed in high vacuum at 1073 K for 1.5 min, followed by quenching in water.



Fig. 2. Virgin magnetization curves and hysteresis loops of  $YCo_5$  (70%)+ $Y_2Co_{17}$  (30%) nanocomposite powders annealed at 1073 K for 1.5 min measured at 3 and 295 K.

They are also very close to the roomtemperature value reported for the singlephase YCo<sub>5</sub>[13,14], which is most likely due to the similar saturation magnetizations of YCo<sub>5</sub> and Y<sub>2</sub>Co<sub>17</sub> [16]. The hysteresis loop measured at 295K reveals a high coercivity value, H<sub>c</sub> = 12.0 kOe and an enhanced remanence,  $\sigma_r/\sigma_{max} = 0.67$ . The coercivity value



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is consistent with the high anisotropy of the YCo<sub>5</sub> phase and the microstructure of this nanocomposite, where the grain size of both phases is in the 10–15 nm range [8,15,17]. The enhanced remanence is a fingerprint of strong exchange interactions between adjacent nanograins. The low-temperature hysteresis loop, T=3K, reveals a greater coercivity value, H<sub>C</sub>=18 kOe, which is 50% more than its room temperature counterpart. This is most likely due to the increase of the YCo<sub>5</sub> anisotropy constant, K<sub>1</sub>, at low temperatures [16]. At the same time, the increase of the squareness value at low T,  $\sigma_r/\sigma_{max} = 0.73$ , might be attributed to the reduction of the thermal effects at 3K.

Fig. 3 shows the temperature dependence of (a) the maximum magnetization  $\sigma_{max}$  and (b) the squareness  $\sigma_r/\sigma_{max}$  of YCo<sub>5</sub> (70%wt)+Y<sub>2</sub>Co<sub>17</sub> (30%wt) nanocomposite powders. The maximum magnetization curve reveals an almost temperature-independent behavior, as the  $\sigma_{max}$  values are within the 80–84 emu/g range. These  $\sigma_{max}$  values are around 70% of the effective saturation magnetization of the nanocomposite ( $\sigma_{s,eff}$  = 120 emu/g). On the other hand, the squareness  $\sigma_r/\sigma_{max}$  increases as the temperature is lowered. As mentioned above, the  $\sigma_r/\sigma_{max}$  increase at lower temperatures can be attributed to the reduction of thermal effects.

A very strong temperature dependence of the coercivity was observed. This is shown in Fig. 4, which illustrates the coercivity HC behavior as a function of temperature T. The most remarkable feature of the  $H_c$  vs. T dependence is that a steeper decrease of the coercivity with increasing temperature occurs at higher temperatures (between 120 and 300 K) than in the 3–120K range. This behavior is in very good agreement with the known T-dependence of the YCo<sub>5</sub>-phase anisotropy constant K<sub>1</sub> [16]. Also worthy of note are the high values of the coercivity at low



temperatures—at 3K, for example,  $H_C = 18$  kOe, some 50% more than its room temperature counterpart.



Fig. 3. (a) Maximum magnetization,  $\sigma_{max}$ , and (b) squareness,  $\sigma_r/\sigma_{max}$ , as a function of temperature *T* for (70%) YCo<sub>5</sub>+(30%) Y<sub>2</sub>Co<sub>17</sub> nanocomposite powders annealed at 1073 K for 1.5 min.



Fig. 4. Coercivity  $H_{\rm C}$  as a function of temperature T for YCo<sub>5</sub> (70% wt)+Y<sub>2</sub>Co<sub>17</sub> (30% wt) nanocomposite powders annealed at 1073 K for 1.5 min.

#### Summary



Nanostructured YCo<sub>5</sub> (70%wt)+Y<sub>2</sub>Co<sub>17</sub> (30%wt) composite powders were prepared by mechanical milling and subsequent annealing at 1073K for 1.5 min. The average grain size <D> of the YCo<sub>5</sub> and Y<sub>2</sub>Co<sub>17</sub> phases was 14 and 12nm, respectively. Temperature-resolved (3–300 K) DC magnetization measurements revealed a strong temperature dependence of both the squareness,  $\sigma_r/\sigma_{max}$ , and the coercivity, H<sub>C</sub>. High coercivity values were observed at all temperatures. The substantial increase of H<sub>C</sub> with decreasing T is attributed to the temperature dependence of the anisotropy constant of the YCo<sub>5</sub> phase. The enhanced remanence ( $\sigma_r/\sigma_{max}$ >0.5) observed throughout the entire temperature range indicates a strong exchange coupling between the hard and soft phases. The maximum magnetization  $\sigma$ max exhibits a very weak temperature dependence. Its value is about 70% of the effective saturation magnetization of the title compound. The observed magnetic properties of the YCo<sub>5</sub> (70%wt)+ Y<sub>2</sub>Co<sub>17</sub> (30%wt) nanocomposite stem from the intrinsic properties of the two phases, as well as from the average grain sizes that favor an effective exchange coupling.

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