Photocatalytic Hydrogen Production under Visible Light over Magnesium Ferrite

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ABSTRACT

Magnesium ferrite (MgFe₂O₄) was synthesized by the hydrothermal technique and was found to be an active photocatalyst for hydrogen production from water under visible light. The structural, morphological, and optical properties of the material were characterized by powder XRD, SEM, TEM, and UV-Vis diffuse reflectance spectroscopy. Iron and magnesium aqueous nitrate solutions were used as precursors under hydrothermal conditions of 200 °C for 3.5 h. MgFe₂O₄ photocatalytic activity towards the H₂ production was determined by gas chromatography, using a batch-type quartz photoreactor and irradiated using a 250 W mercury lamp. XRD results from the synthesized sample found the MgFe₂O₄ crystalline structure. The optical properties revealed semiconducting properties with a band gap energy (E_g) of 2.07 eV (599nm) showing an efficient visible light absorption. SEM images found particles with a morphology in the form of agglomerates composed of hemispherical particles, while TEM images revealed particles with an average of 9.3 nm in size. Furthermore, the solid exhibited a high photoactivity toward the reduction of water, which is attributed to the efficient separation and transportation of the photogenerated charge carriers. This ferrite material exhibited a production of 650 µmol H₂/g_{catalyst} over an irradiation period of 8 h, thus exceeding the H₂ generation obtained by TiO₂, which was of 15 µmol H₂/g_{catalyst}. This increase in H₂ production is attributed to the fact that MgFe₂O₄ exhibits a band gap, which is activated under the visible light range.

Keywords: Hydrogen production, MgFe2O4, Water splitting





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1. Introduction

In the last decade it has been an increased research interest related to renewable energy, because of environmental constraints and the low efficiency offered by traditional fossil energies. It is thought that the development of a new sources of renewable energy will bring a number of advantages [1]. A promising alternative is the use of hydrogen, because it has so many desirable features such as clean emissions when used as fuel, abundance, accessibility, high efficiency and overall versatility of use [2]. However, the first step in the production of a safe and clean energy, is finding a method for producing hydrogen cheaply and efficiently [1]. Currently, hydrogen is proposed as an energy vector that can be applied to renewable energy source (RES) such as solar, wind, biological, ocean and geothermal. Of these alternative RES it is important to take in to account factors like accessibility, affordability and being environmentally friendly. Based on this the use of water can be considered as an ideal source of hydrogen [3, 4, 5, 6, 7].

In the literature there are reports of hydrogen production from water photo reduction in the presence of semiconductors (SC) as photocatalysts. That is why the exploration of new semiconductors with high photocatalytic activity under visible light irradiation, projected various advantages such as low cost, favorable positioning of the band energy levels and chemical stability in aqueous media [8].

Ferrites with the general formula MFe_2O_4 , where (M) represents a metal cation have various applications because of their chemical and thermal stability. Their applications mainly has excelled in electronic devices and circuits, besides their increasing use as absorbers of toxic and hazardous substances to health. Ferrites have shown to be effective photocatalysts under visible light to generate electron-hole pairs on the photocatalytic material surface. However, there is still a great challenge in understanding the mechanisms of water reduction [9].

Magnesium Ferrite (MgFe₂O₄) has a band gap energy of 2.18 eV [9], indicating that it can be activated under the visible light range. Literature reports several synthesis methods for preparing the magnesium ferrite spinel, such as ceramic [10-12], hydrothermal [13], sol-gel [14], citrate gel [15], combustion [16 - 17], mechanical alloying [18], coprecipitation [19], etc. However, the coprecipitation method has some advantages such as: simple, fast, easy control of particle size and composition and various possibilities for controlling the state of the total surface of the particles and their homogeneity. Generally, after performing coprecipitation, the material is exposed to a heat treatment to crystallize, but runs the risk of particle growth and then exceed the nanometer scale. Therefore, in the present work the precipitation is to be combined with hydrothermal treatment to obtain a material with all the above mentioned advantages of the hydrothermal treatment.

The objective of this work is to synthesize magnesium ferrite by the coprecipitation combined with hydrothermal treatment, characterize their structural, morphological, textural and optical properties, and to evaluate its photocatalytic activity towards the hydrogen production under visible light.









2. Experimental

2.1. Synthesis of nanoparticles

Precursor solutions were synthesized by the coprecipitation/hydrothermal method, using precursors: Magnesium II nitrate (Mg(NO₃)₂*4H₂O) Sigma Aldrich brand with a 98% purity and Iron III nitrate (Fe (NO₃)₃*9H₂O) Sigma Aldrich brand with a purity of 98%. The material was prepared in order to obtain 1 g of MgFe₂O₄, for that 4.39 g of Fe(NO₃)₃*9H₂O was added in 5 ml of water and 1.39 g of Mg(NO₃)₂ in 3 ml of water. These solutions were mixed and placed in a burette and allowed to drip into 20 ml of ammonium hydroxide (NH₄OH) as a precipitating agent and pH regulated to a value of approximately 10. Subsequently, the mixture was exposed to a hydrothermal treatment in an autoclave using a Teflon vial at 200 °C for 3 h.

2.2. Material Characterization

 $MgFe_2O_4$ powders were characterized by different techniques in order to study its physical and chemical characteristics. In order to determine the crystal structure of the material, this was analyzed by XRD using a Bruker D8 Advance model X-ray diffractometer with a 1.54060Å wavelength and employing 15 to 85 ° in 20 with a step-size of 0.05 and 120s by step. The crystal size was estimated using the Scherrer's equation. The light absorption spectra of the material was acquired by UV/Vis diffuse reflectance spectra using a Perkin Elmer (Lambda-10) spectrometer equipped with an integrating sphere. The photocatalyst morphology was examined by transmission electron microscopy (TEM), in addition the particle size thereof was measured by a particle counting on a JEM-2200FS transmission electron microscope of ultra-high resolution that combines an emission gun of 200kV with an omega filter. The surface area of the powders was evaluated by the nitrogen physisorption technique (BET) to a degassing temperature of 180 °C, using a Quantachrome Autosorb-1 BET aparatus.

2.3. Photocatalytic Evaluation

200 mg of $MgFe_2O_4$ nanoparticles were suspended in water and methanol, the latter was used as a sacrificial agent; the suspension was placed inside a quartz batch type photoreactor (Figure 1) with a length of 19 cm and a diameter of 5 cm, which was sealed and located 7 cm away from a 250W mercury lamp. Hydrogen production was monitored using a gas chromatograph Perkin Elmer Clarus 500. Photocatalytic activity was evaluated over a period of time of 8 hours with sampling intervals of every hour.

Since titanium dioxide (TiO_2) is a reference photocatalyst, TiO_2 was synthesized by the sol-gel method assisted with hydrothermal treatment (TiO_2-F) , both physical and chemical properties as well as the photocatalytic activity towards the hydrogen production were compared via the dissociation of the water molecule.











Fig 1. Diagram for the photocatalytic evaluation process

3. Results and discussion

3.1. Difracción de Rayos X

In Figure 2 the X-ray diffractograms of the MgFe₂O₄ and TiO₂-F photocatalysts are shown. From them the diffraction angles of the crystalline structures present in the materials can be observed. For the MgFe₂O₄ sample a mixture of phases was observed; α -Fe₂O₃ and the spinel MgFe₂O₄. Observing the diffraction angles of 33.4 °, 42.2 °, 57.5 °, 63 ° which correspond to the reflections (311), (400), (511) and (440) of cubic MgFe₂O₄. While angles 24.3 °, 33.1 °, 40.8 °, 49.3 °, 54.1 ° correspond to reflections (012), (104), (113), (024) and (116) that identify the phase α -hematite Fe₂O₃ [20]. The result of the phase mixture can be explained since the phase diagram Fe₂O₃-MgO-O₂ (P, atm) indicates that at stoichiometric ratios and low temperatures a single-phase mixture is reached as presently obtained.

For the case of TiO₂-F peaks located at 25.4, 37.8, 48.0, 54.5 $^{\circ}$ of 2 Θ the corresponding to planes (101), (004), (200), (105 and 211) of the anatase phase are observed [22].



Fig 2. MgFe₂O₄ and TiO₂-F sample diffraction patterns.









Comparing the width of the peaks of the TiO_2 -F and $MgFe_2O_4$ and sample it can be observed that narrower signals for the ferrite, suggesting a greater crystal size for this sample. To determine the crystal size of the samples X-ray patterns data (characteristic peak of each material) and the Scherrer's equation was employed, thus TiO_2 -F presented a crystal size of 10.5 nm, while $MgFe_2O_4$ showed a size of 44 nm, which proves the earlier assumption about the width of the diffraction patterns peaks. This difference in crystal size can be attributed to the different synthesis methods used in each material (different mechanisms of nucleation and growth), coprecipitation for ferrite and sol-gel for the titanium oxide.

3.2. UV-Vis Spectroscopy

UV-Vis diffuse reflectance spectra for $MgFe_2O_4$ and doped TiO₂-F converted to Kubelka Munk units are presented in Figure 3. To determine the sample band gap it was considered a linear trend from inflection region of the spectrum, which represents the energy absorption over the edge. Extrapolating the slope of the linear region to intercept the photon energy axis (x-axis), this point provides the value of optical band gap of the material.

In this Figure it can be seen that the value of energy band gap (Eg) for the TiO₂-F sample is 3.12eV, which is very close to the reported value for the TiO₂ anatase phase of 3.2 eV [22]. On the other hand sample MgFe₂O₄ presents an Eg value of 2.07eV. This value is within the values reported for magnesium ferrite spinel which are between 2.0 and 2.18 eV, and these vary depending on the particle size; the smaller the value of the band gap this increases [9].



Fig 3. Band gap determination by UV-Vis spectroscopy.









3.3. Transmission electron Microscopy (TEM)

Transmission electron microscopy images for samples $MgFe_2O_4$ and TiO_2 -F are presented in Figure 4. In this figure it can be seen completely different morphologies; $MgFe_2O_4$ particles present are cubic shapes with a mean size of 90 nm. While for the TiO_2 -F sample ellipsoidal particles with an average size of 20 nm can be observed. These results, as in the case of the crystal size calculations, are to be expected as various methods of synthesis were used, which correspond to different mechanisms of nucleation and particle growth.



Fig 4. TEM images for the samples MgFe₂O₄ and TiO₂-F

3.4. Nitrogen Physisorption (BET)

The method of nitrogen physisorption (BET) was employed for obtaining the surface area of the synthesized materials (MgFe₂O₄ and TiO₂-F). Figure 5 shows the adsorption isotherms, where it appears that TiO₂-F isotherm is of type IV according to the classification of the IUPAC, which is characteristic of mesoporous materials, with a H2 hysteresis loop characteristic bottleneck type pores. However, the adsorption isotherm for MgFe₂O₄ is of type III characteristic of a hysteresis loop H3 of mesoporous materials with a very wide size distribution [21], which can be associated with interparticle porosity, very characteristic of materials obtained by precipitation.

Moreover, in Figure 5 it can be observed that there is a considerable difference in the amount of adsorbed nitrogen, being higher for the sample TiO_2 -F, this is attributed to presenting greater surface area. This result was expected since by size particle (small) as well as by porosity high surface areas are generated. The value of the specific surface area and the particle size obtained by TEM images for each material is presented in Table 1.





Fig 5. MgFe₂O₄ and TiO₂-F adsorption isotherms.

Table 1. BET surface area and particle size for the synthesized photocatalysts

Sample	Area (m ₂ /g)	Particle size (nm)
TiO ₂ -F	124	20
MgFe ₂ O ₄	53	90

3.5. Photocatalytic Evaluation

In Figure 6 the hydrogen evolution of both MgFe₂O₄ and TiO₂-F samples are presented. It is evident that an increased hydrogen production for sample MgFe₂O₄ was achieved, although it has less surface area than the TiO₂-F, the photocatalytic activity mainly is attributed to the lower value of Eg, which is in the range of visible light as mentioned in Figure 3. More than 650 μ molH₂/g_{cat} was reached for sample MgFe₂O₄ after 8 h of irradiation against 15 μ molH₂/g_{cat} produced by the synthesized TiO₂-F.





Fig 6. Photocatalytic evaluation for the production of H₂

4. Summary and perspectives

Magnesium ferrite was synthesized by coprecipitation and hydrothermal treatment, resulting in a mixture of phases (MgFe₂O₄ and Fe₂O₃), while TiO₂-F sample was obtained by sol-gel assisted with hydrothermal treatment, reaching the pure anatase phase. Results obtained by diffuse reflectance UV-Vis spectroscopy showed the Eg value for the ferrite was located in the visible region, 2.07 eV (599nm). TEM results show particle sizes in the nanometer range in both cases, and also indicated a cubic morphology for MgFe₂O₄, while ellipsoidal particles for TiO₂-F. The photocatalytic evaluation results for the H₂ production indicate a higher activity for the MgFe₂O₄ sample, with a maximum production of 650 μ molH₂/g_{cat} over a period of 8, while TiO₂-F only achieved a production of 15 μ molH₂/g_{cat}. This is due to the fact that magnesium ferrite is able to be activated under the visible light range.

Acknowledgements

Finally, thanks to technicians Maria Teresa Chaparro, Enrique Torres, Luis de la Torre and Carlos Ornelas for their contributions to the of UV-Vis spectroscopy, BET area, X-ray diffraction and scanning electron microscopy results. As well as the Centro de Investigación en Materiales Avanzados S. C. for the support in the use of its infrastructure.



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