

Spin-Lattice coupling, Jahn-Teller effect and the influence of the measurement rate in $\text{La}_{0.7}\text{Ca}_{0.3-x}\text{Sr}_x\text{MnO}_3$ manganites

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Abstract

This work presents a study of how the magnetic and temperature variables, as well as the measurement rate, affect magnetic and structural phase transitions for $\text{La}_{0.7}\text{Ca}_{0.3-x}\text{Sr}_x\text{MnO}_3$ ($x=0.15, 0.08, 0.07$ and 0.06) manganites. A single phase rhombohedral crystal structure for $\text{Sr}_{0.15}$ compound and a single phase orthorhombic crystal structure for $\text{Sr}_{0.08}$, $\text{Sr}_{0.07}$, and $\text{Sr}_{0.06}$ compounds were identified by X ray diffraction. The Curie temperature adjustment to room temperature was reached for $\text{Sr}_{0.07}$ sample, in which a Jahn-Teller transition was observed. This Jahn-Teller transition showed a susceptible response to 0.5, 1, 2, 3, 5 and 7 K/min measurement rates, and was revealed without measurement sequence at 234, 257, 196, 233, 245 and 238 K respectively in thermomagnetic curves (10mT). Also, Curie temperature increases sequentially around 2 K for the same measurement rates. To avoid magnetic field effect, a differential scanning calorimetry for $\text{Sr}_{0.07}$ sample was obtained in the same temperature region, for 2, 3, 5, 7 K/min showing the Jahn-Teller phase transition at 225 K and a vibrational like behavior in the heat flux below 225 K, for all the temperature rates. The latter, is related to the static to dynamic Jahn-Teller transition. Furthermore, Curie temperature appears around 284 K.

Introduction

The strong spin-lattice coupling in manganites has evidenced a large influence in magnetocaloric properties and phase transition behavior arisen in them. This mixed valence manganites with general formula $R_{1-x}A_xMnO_3$ (where R is a trivalent rare-earth element: La, Pr, Nd, Sm etc. and A is a divalent or monovalent ion Sr, Ca, Ba, Na, K etc.) are promising materials for magnetic refrigeration applications near room temperature. They become ferromagnetic and metallic by doping with monovalent or divalent cations, making enough Mn^{3+} and Mn^{4+} ionic pairs ($Mn^{3+}/Mn^{4+} \approx 7:3$) which interact by double exchange.[1] The magnetocaloric properties observed in these manganites have high susceptibility to ionic radius, dopant valence and fabrication method; because the spin, lattice, charge and orbital degrees of freedom are related with the crystalline structure.[1, 2] The double exchange mechanism has been argued as responsible for the magnetic structure and electronic transport properties of these compounds. However a strong Jahn-Teller (JT) electron phonon coupling arising from the $Mn^{3+}O_6$ octahedron due to JT effect should play an important role.[3] The divalent cations distort the structure due to radius differences, leading to the Jahn-Teller effect. The deformation of the MnO_6 octahedron originates from two causes: the Jahn-Teller effect with double degeneracy of e_g orbitals and from the connection pattern of the MnO_6 octahedron in the perovskite structure (tilting of the octahedra), which forms rhombohedral or orthorhombic lattices.4 Spin-lattice coupling is reflected in several macroscopic properties, such as in strong lattice contraction upon cooling through Curie temperature (T_C), large or anomalous magnetostriction of some manganites, associated structural and magnetic phase transitions and strong dependence of T_C on epitaxial

strain films.[5] Next, this work explains the influence of magnetization and the heat flux variables measured at different temperature rates in order to adjust Curie temperature with doping and to investigate the spin-lattice coupling behavior in manganites.

Experimental

The compositions from $\text{La}_{0.7}\text{Ca}_{0.3-x}\text{Sr}_x\text{MnO}_3$ system were fabricated using a solid state reaction method at high temperature. Precursor powders of La_2O_3 , Mn_2O_3 , CaCO_3 and SrCO_3 were mixed in stoichiometric proportions according to $x= 0.06, 0.07, 0.08$ and 0.15 . Using a pulverisette 7 low energy planetary mill, the samples were ground initially for 4 h and then milled also for 4 h between heat treatments. The samples were presintered at $750\text{ }^\circ\text{C}$ for 15 h and at $950\text{ }^\circ\text{C}$ for 18 h in air atmosphere. Finally, the sample powders were pressed into a die of 1 cm of diameter and subsequently sintered at $1300\text{ }^\circ\text{C}$ for 24 h. X ray powder diffraction patterns were measured at room temperature in a Panalytical diffractometer model X'pert Pro-MPD with Cu K α radiation ($\lambda=1.5406\text{ \AA}$). The calorimetric measurements were performed in a TA instruments model DSC Q200 calorimeter. The measurement of thermomagnetic curves were carried out in a Quantum Design Physical Properties Measurement System (PPMS). Raman measurements were made in a Raman Horiba model LabRam HR VIS 633 with He-Pe laser of 632 nm.

Results and discussion

Crystalline phases were identified by X-ray diffraction for $x=0.15, 0.08, 0.07$ and 0.06 samples in $\text{La}_{0.7}\text{Ca}_{0.3-x}\text{Sr}_x\text{MnO}_3$ system as shown in figure 1. According to crystallographic file PDF 01-071-5287 a single phase for $\text{Sr}=0.15$ sample with a rhombohedral structure and a R-3c space group was identified. Moreover, for $\text{Sr}= 0.08$,

0.07 and 0.06 samples a single phase with orthorhombic structure and Pnma space group were identified according to crystallographic file PDF 01-071-5292 for these samples.

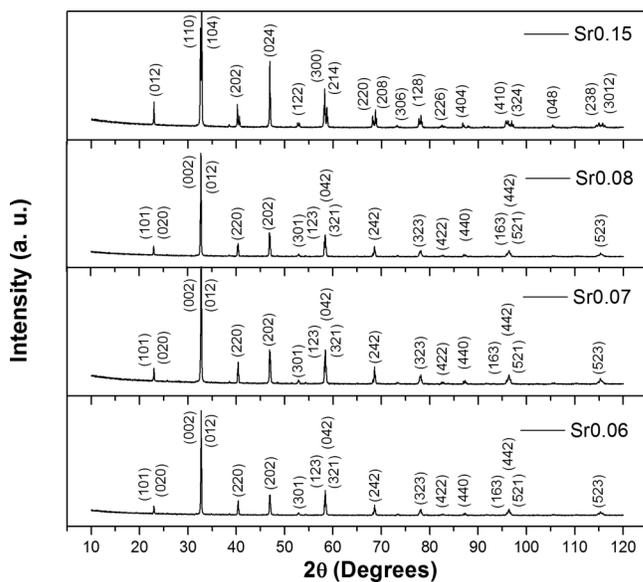


FIG. 1. X ray diffraction patterns for Sr = 0.15, 0.08, 0.07, 0.06 powders samples in $\text{La}_{0.7}\text{Ca}_{0.3-x}\text{Sr}_x\text{MnO}_3$ system.

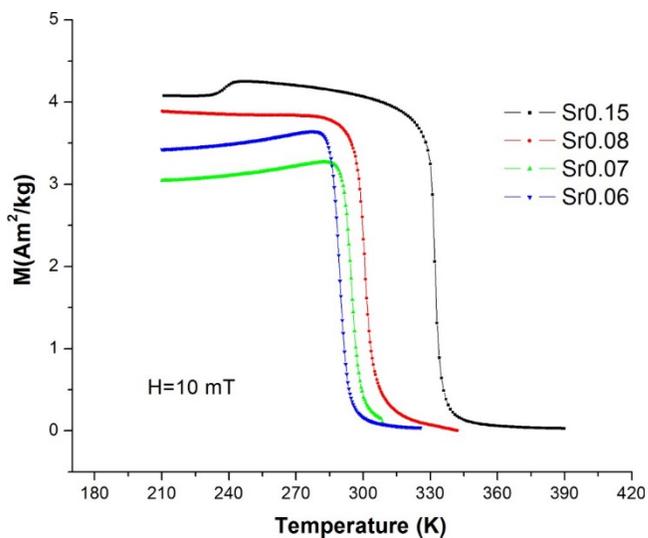


FIG. 2. M vs T curves for Sr = 0.15, 0.08, 0.07 and 0.06 samples, measured at 15 K/min under a constant magnetic field of 10 mT.

Thermomagnetic curves (M vs T) were measured at 15 K/min, under a constant magnetic field of 10 mT, for $x = 0.15, 0.08, 0.07$ and 0.06 samples in order to observe the magnetic transition temperature (see Fig. 2). The entire samples pass through a second order [6] ferromagnetic to paramagnetic phase transition, and a displacement of the Curie temperature with doping was noticed. The Curie temperatures for $Sr = 0.15, 0.08, 0.07$ and 0.06 samples are 332, 300, 295 and 290 K respectively. The Curie temperature was adjusted very close to room temperature as each 0.01 of Sr increase in stoichiometric formula generates a Curie temperature raise of around 5 K. The small shoulder located at 238 K in $Sr = 0.15$ thermomagnetic curve corresponds to an orthorhombic to rhombohedral phase transition. [7]

In order to identify ferromagnetic to paramagnetic phase transition behavior when a measurement temperature rate is applied; M vs T curves were obtained for orthorhombic $Sr_{0.07}$ sample at measurement rates of 0.5, 1, 2, 3, 5 and 7 K/min as shown in figure 3. The Curie temperature changes as follows: 286.80, 287.28, 288.44, 289.36, 291.48, and 293.43 K, as the measurement rate increases for 0.5, 1, 2, 3, 5 and 7 K/min respectively. In the low temperature region (196 – 257 K) another transition is observed. This transition shows a sensitive response to 0.5, 1, 2, 3, 5 and 7 K/min measurement rates, because occurs without sequence when temperature rate is increased at 234, 257, 196, 233, 245 and 238 K respectively. Therefore, the influence of the temperature rate as well as the applied magnetic field is detected in this system with strong spin-lattice coupling.

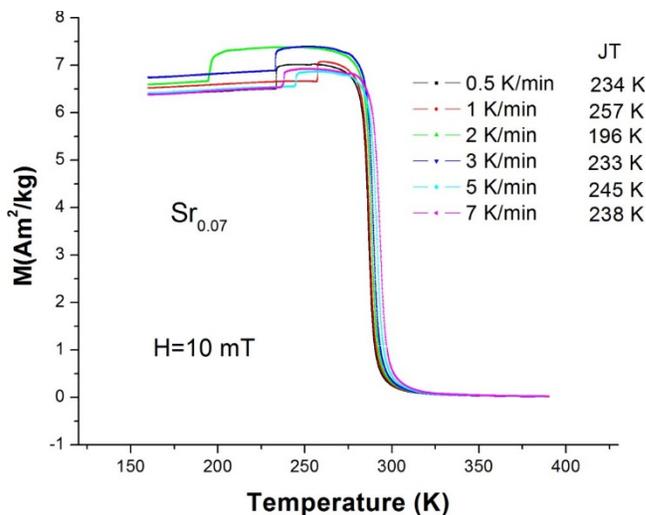


FIG. 3. M vs T curves for $\text{La}_{0.7}\text{Ca}_{0.23}\text{Sr}_{0.07}\text{MnO}_3$ sample measured at 10 mT for 0.5, 1, 2, 3, 5 and 7 K/min measurement rates.

Given the distorted nature of orthorhombic structure and thinking about vibrational behavior according to different responses of measurement temperature rates in figure 3, a Raman spectrum was acquired. Figure 4 shows a Raman spectrum measured for the $\text{La}_{0.7}\text{Ca}_{0.3-x}\text{Sr}_x\text{MnO}_3$ manganite inside a spectral range of 0 to 2000 cm^{-1} . This spectrum shows two main vibrational modes, one observed at 210 cm^{-1} ascribed to a tilt mode of MnO_6 octaedra [8] and other observed at 480 cm^{-1} that corresponds to a Jahn-Teller mode. [9] This Jahn-Teller mode occurs because the MnO_6 octahedron environment requires an energy balance. The higher energy e_g electron in Mn^{3+} ion decreases electrostatic (repulsive) energy by an increase in the elastic energy, which implies the e_g orbital splitting and therefore a MnO_6 octahedron distortion. [9]

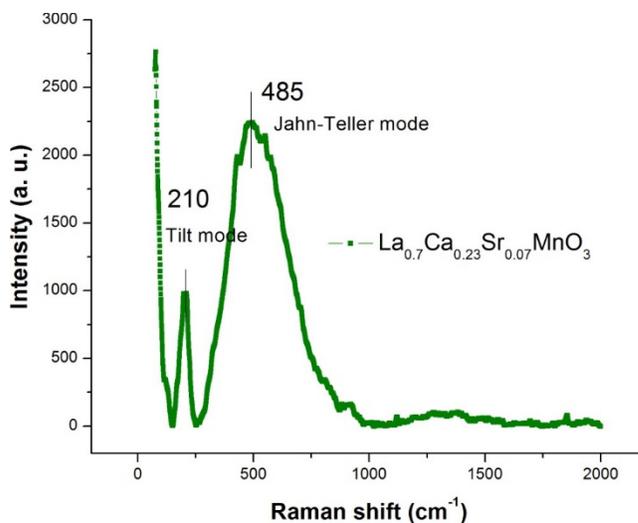


FIG. 4. Raman spectrum for $\text{La}_{0.7}\text{Ca}_{0.23}\text{Sr}_{0.07}\text{MnO}_3$ sample.

These vibrational modes identified by Raman spectroscopy justify the presence of the JT transition without measurement rate sequence, observed at low temperatures (196 – 257 K) in M vs T curves for $\text{Sr}_{0.07}$ sample. The temperature rates and the applied magnetic field stimulate the vibrational modes of the orthorhombic lattice in $\text{Sr}_{0.07}$ sample because when carriers have certain mobility, the distribution of Mn^{3+} and Mn^{4+} ions is random and changes with time. [4]

Aiming to explain this temperature transition behavior at low temperatures (196 – 257 K) in M vs T curves and due the high spin-lattice coupling in these compounds a differential scanning calorimetry (DSC) was measured to avoid the magnetic field influence.

The DSC obtained for $\text{Sr}_{0.07}$ sample was measured at 2, 3, 5 and 7 K/min in cooling and heating regimes inside a temperature range of 300 to 203 K for cooling and from 203 to 320 K for heating (see Fig. 5). In DSC graph two transitions were observed for each measurement rate used, both in cooling and heating regimes. Analyzing the

heating rate regime, the endothermic peak observed around 284 K is related with the ferromagnetic-metal (FMM) to paramagnetic-insulating (PMI) transition, occurred at metal-insulator transition temperature ($T_{MI} \sim T_C$). For every doping level in $\text{La}_{0.7}\text{Ca}_{0.3-x}\text{Sr}_x\text{MnO}_3$ system two mainly states appear, FMM and PMI according to phase diagram. [10] The double exchange interaction in this compounds is related to electron hopping between Mn^{3+} and Mn^{4+} ions through Oxygen atom which generates a ferromagnetic-metal state from a paramagnetic-insulated one. The conduction band is narrow above T_C in the paramagnetic- insulating state leading to electron localization. [11] In this paramagnetic-insulating state charge transport is thermally activated for hopping around T_{MI} transition temperature, which means that e_g electrons of Mn^{3+} ions are mainly localized before the transition and they should overcome and energy barrier to hop to Mn^{4+} ions through the oxygen atom. [5, 12] The latter explain the transition around T_C in figure 5.

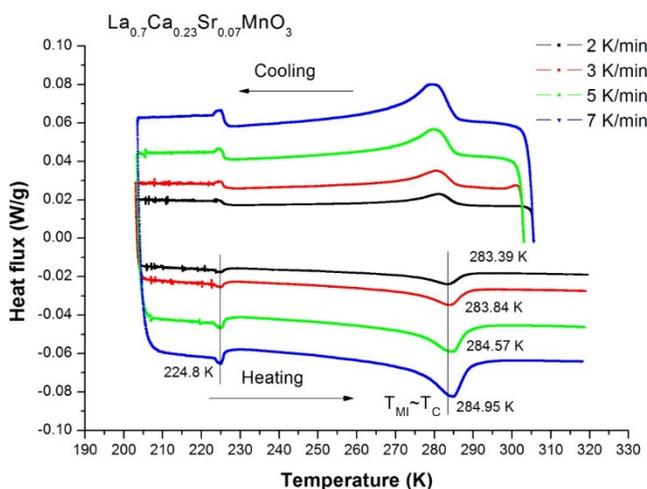


FIG. 5. Differential scanning calorimetry for $\text{La}_{0.7}\text{Ca}_{0.23}\text{Sr}_{0.07}\text{MnO}_3$ sample in cooling and heating regimes at 2, 3, 5 and 7 K/min measurement rates.

The endothermic peak occurred at 225 K and the heat flux below this temperature, shown in figure 5, appears for all measurement rates in heating and cooling regimes. These are attributed to the JT and tilt modes identified by Raman. The oscillating flux is attributed to the phase transition from a JT static cooperative state above 225 K to a dynamic JT state below [13] 225 K. In static cooperative JT phase the distortions are correlated; identically distorted molecules establish a particular orbital array that allows e_g electrons to move through these orbitals preserving the initial phase distortion. [14, 15] Whereas in dynamic JT phase the distortions are not correlated, establishing a random phase where the molecules perform random motions and the electrons move according to their changing orbital surroundings. [14]

The endothermic peak shown at 225 K for all measurement temperature rates in figure 5, and the transitions observed between 196 and 257 K in figure 3 belong to the same second order [16] JT transition. The behavior change between these two measurements could be related with the fact that magnetic field enhances metallic behavior due to double exchange interaction. [17] When temperature rates change in this enhanced FMM state, the amount of energy applied with each temperature rate activates different vibrational modes of the orthorhombic structure due to random orientation of Mn^{3+} and Mn^{4+} ions, triggering the JT transition between 196 and 257 K.

The static JT cooperative distortion is possible in the orthorhombic crystal structure ($x=0.07$) however the higher symmetry of MnO_6 octahedron in the romboedral phase ($x=0.15$) with a single Mn-O bond length does not allow it. The JT distortion modes are tetragonal or orthorhombic but the romboedral symmetry does not split the e_g orbital. [6]

Conclusions

As presented here, the Curie temperature could be adjusted to room temperature in $\text{La}_{0.7}\text{Ca}_{0.3-x}\text{Sr}_x\text{MnO}_3$ using a fine Sr doping and the methodology proposed. The presence of Jahn-Teller vibration modes and MnO_6 octahedron tilt were identified by Raman spectroscopy in $\text{Sr}_{0.07}$ sample. These modes appear at 480 and 210 cm^{-1} respectively and are observed in DSC and M vs T graphs as a Jahn-Teller transition. The appearance of JT transition for temperatures between 196 and 257 K in M vs T curves is due the strong spin-lattice coupling of this system and to the occurrence of the static to dynamic Jahn-Teller transition. The static to dynamic JT transition is revealed in DSC graph with an endothermic peak at 225 K which appears independently of temperature measurement rate. Also, the dynamic JT phase is evidenced with thermal fluctuations occurred for temperatures below 225 K. The factors shown here must be considered for the experimental measurements and potential practical use either for manganites or any other family of materials intended to be studied as a magnetocaloric material.

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